1	Feasibility of embedded distributed optical fibre sensors in thermoplastic
2	composite braided beam structure
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11 Abstract

12 In this paper, consolidation monitoring and structural health monitoring (SHM) of a braided 13 thermoplastic composite beam was performed using a distributed optical fibre sensor (DOFS). The 14 DOFS was manually embedded into the braided preform before melt processing. The real-time strain 15 and temperature data obtained during consolidation and cooling segments were correlated with phase 16 transitions in the thermoplastic matrix. Post consolidation, the quality of the embedded DOFS was 17 investigated using a micro-CT to reveal defects such as waviness and misorientation due to the crude 18 nature of the adopted embedding technique. The manufactured beam was then subjected to repeated 19 loading-unloading cycles in a flexure test. The strain developments along the embedded optical fibre 20 length were comparable with the measurements from Digital Image Correlation (DIC), and further 21 correlated with the post damage observations. Despite the crude method of embedding the optical fibre, 22 the monitored data was useful for consolidation monitoring as well as SHM. This proved that DOFS 23 could be embedded into composite structures without adding no cost, time and complexity thus making 24 them feasible for industrial applications.

Keywords: Distributed Optical Fibre Sensor (DOFS); Braided composites; Consolidation monitoring;
 Structural Health Monitoring (SHM); Micro-CT.

1 1 Introduction

Fibre-reinforced polymer composites remain at the heart of future plans for lightweight structures in sectors as diverse as aerospace, road/rail-based passenger vehicles and offshore power generation. This is due to their high strength-to-weight ratio, as well as design flexibility and corrosion resistance and durability. A significant number of composite manufacturing processes have been developed during the last 40 years including compression moulding, vacuum/bag/autoclave moulding, resin transfer mounding (RTM), filament winding etc. [1], depending on the material, design and the application.

8 A critical disadvantage of composites is the occurrence of barely visible damage (BVD) [2–4]. Given 9 the complex multi-layered structure of composites, these defects are extremely difficult to detect, but 10 bear the potential to significantly compromise the structural properties. Consequently, designers are 11 compelled to use conservative factors of safety, which restrict the maximum exploitation of the unique 12 properties possessed by composites. In terms of structural inspections, current industry practice involves 13 scheduled maintenance that involves routine but rigorous checks for such hidden damages. Such 14 inspections often prove trivial but require the structures to be taken out of service, resulting in significant 15 loss of revenue. In the aerospace industry, inspection accounts for 27% of the total lifecycle cost of an aircraft [5]. Therefore, there have been significant efforts aimed at replacing scheduled maintenance 16 17 with condition-based maintenance [6], enabled by active Structural Health Monitoring (SHM) 18 technologies.

19 A preferable approach which has received some attention is in-situ internal health monitoring which 20 typically relies on multiple sensors installed across the entire structure that provide regular status of the 21 structure's safety. However, typically available sensor technologies provide point-based 22 strain/temperature measurements and large-scale monitoring would add significant cost, complexity 23 and mass. The embedding of optical fibre sensors does not have a significant impact on the mechanical 24 properties of the structures when used for in-situ composite monitoring [7–9]. This has been 25 demonstrated with the traditional fibre optic sensor, i.e., Fibre Bragg Gratings (FBG), which are used 26 to conduct measurements at single or discrete points along the length of an optical fibre. A more recent 27 technology, i.e., Distributed Optical Fibre Sensors (DOFS) provide further advantage by eliminating 28 the need to etch individual gratings into a fibre, hence enabling a continuous measurement along the 29 optical fibre at a much lower cost. Several authors have investigated the potential of embedding 30 different types of optical fibres into composite structures for real-time structural health monitoring to 31 detect cracks, delamination or debonding by relating the distortion spectrum to the specific defect and 32 damaging mechanism. Moreover, a few notable studies have also investigated the potential of optical 33 fibres for monitoring composites during the manufacturing process (curing for thermosets; 34 melting/reconsolidation for thermoplastics) and capture the phase transitions at material level. Table 1 35 summarises such investigations from the existing academic literature.

Table 1 The usage of optical fibre technique in the structural health monitoring and process monitoring of composites reported in the literature

Reference	Materials	Type of optical fibre	Structural health monitoring	Process monitoring
Rufai et al. [9,10]	Woven E-glass fibre fabric and epoxy resin composite plate by Vacuum- Assisted Resin Infusion Moulding (VARIM)	Micro-braided DOFS	Four-point bending test	Residual strain measurement in the longitudinal and transverse directions
Mulle et al. [11]	Glass-fibre-reinforced polypropylene (GFRP)	FBG		Strain variations related to physical change of the laminate related with cooling rate and coefficients of thermal expansion (CTE)
Sorensen et al. [12]	Carbon fibre-reinforced polyphenylene sulphide (AS4/PPS) thermoplastic composite	FBG		Strain development responded to changes in material state, especially the glass-rubber transition and solid–liquid transition
Arhant et al. [13]	Unidirectional C/PA6 thermoplastic panel	a shamrock shaped DOFS placement		Strain measurement of along all the principal directions $(0, \pm 45 \text{ and } 90^{\circ})$
Huang C et al. [14]	Carbon fibre/Epoxy by autoclave	Brillouin-scattering- based DOFS		Residual strain measurement
De Baere et al. [15]	Carbon fabric-reinforced PPS	FBG	Fatigue load measurement and the FBG showed the capability of surviving over half a million loading cycles without debonding	
Califano et al. [16]	Unidirectional carbon fibre fabric	Polyimide-coated silica glass DOFS	Integrate with machine learning methods to associate to damage patterns	
Sawicki et al. [17]	Ultra-High Performance Fibre Reinforced Cemetitious (UHPFRC)	DOFS	Four-point bending test of a UHPFRC beam to precisely detect and localize the microcracks in the elastic phase	
Lau et al. [18]	Composite-strengthened concrete	FBG	Strain measurement when the structure is subjected to debond or micro-crack failure	

Ryu et al. [19]	CFRP	FEG	Instrumented a wing box using 24 FBG sensors and an interrogator based on a wavelength-swept fibre laser (WSFL) demonstrating the capability to monitor in real- time the buckling behaviour of a wing box.	
Minakuchi et al. [20]	CFRP panel	DOFS	Estimate effectively the the damage which occurred during impact tests	Estimate effectively the residual strains after manufacturing

1 To the best of authors' knowledge, these published works applied different kinds of optical fibres in flat 2 thermoset/thermoplastic coupons or panels made by tradition manufacturing tools. Composites 3 manufactured via braiding have attracted significant interest in a variety of industries such as 4 automotive, aerospace and battery enclosures for advanced electric vertical take-off and landing 5 vehicles [21], due to the advantages that the shape of the resulting braiding preforms could be matched 6 closely with that of the finally required components, thus reducing material wastage [22]. Moreover, 7 the interlacement of the fibres imparts superior resistance to damage propagation compared to 8 traditional laminates [23].

9 Another noteworthy observation from the reviewed papers is the use of localisation mechanisms, for 10 instance, epoxy adhesives in [9], to ensure consistent and controlled orientation of the optical fibres 11 within the composite's structures. This is because the shift between the yarns and warp during the 12 laying-down of optical fibre is difficult to control. A few investigations presented the use of advanced 13 non-destructive techniques such as micro-CT [24] to interpret the position profile or quality control of 14 a FBG when inserted into a carbon polyphenyl sulphide (PPS) woven fabric plates. While controlled 15 orientation of the optical fibres is deemed important for understanding material behaviour, the practices 16 implemented to maintain orientation may reduce the practical feasibility of these systems by adding 17 cost and/or time to the manufacturing process, especially for woven and braided materials. Therefore, 18 it is important to question the importance of orientation, particularly for SHM, as the relative change in 19 the measured signal irrespective of orientation could prove sufficient for the intended purpose.

20 Considering the research gaps above, this paper presents an investigation of process as well structural 21 health monitoring for a tubular braided thermoplastic composite beam using an embedded DOFS 22 system. Compared to previous studies, the DOFS was embedded during the manufacturing of a braided 23 GFRP beam with relatively less control for ensuring minimum disruption of the process. Following the 24 manufacturing process, the position check and quality control of the embedded DOFS were further 25 interpreted by a micro-CT investigation. Strain data was continuously acquired from the DOFS and 26 processed to perform consolidation monitoring during the moulding process. Furthermore, structural 27 health monitoring during mechanical loading cycles applied to the manufactured part. The results of the 28 study indicate if the DOFS system is robust enough to replicate its ability to identify damage initiation 29 without controlled orientation

30 2 Rayleigh scattering based DOFS

Herein we examine Rayleigh scattering based optical fibre sensors due to their high chemical stability and small diameters. Such fibres were embedded to the anode electrode of a prototype pouch cell to monitor the distributed strain and temperature in real-time during various operating conditions.

When an electromagnetic wave is launched into an optical fibre, the light will be redistributed byRayleigh scattering [25]. If the local changes in temperature and strain are relayed to the optical fibres,

the scattered signal in the fibre will be modulated by these physical parameters. As a result, by measuring changes in the modulated signal, it is possible to quantify local physical variables of temperature and strain. The desired features of Rayleigh scattering based fibre sensing technique allows distributed measurements with millimetre-scale spatial resolution and high measurement accuracy, making it a suitable solution for in-situ applications.

6 Coherent Optical Frequency Domain Reflectometry (C-OFDR), a type of Rayleigh scattering based 7 DOFS, is performed to monitor distributed strain and temperature of the anode electrode. C-OFDR was 8 selected due to its high spatial resolution (2.6 mm) measurement capability. The configuration consists 9 of a main interferometer and an auxiliary interferometer, as shown in supplementary Figure 1. The main 10 interferometer contains two optical couplers (OC1 and OC2), an optical circulator (CIR), two 11 polarization controllers (PC1 and PC2), a fibre under test (FUT) and a polarization beam splitter (PBS). 12 The auxiliary interferometer includes two optical couplers (OC3 and OC4). The optical frequency of 13 the output laser from the tuneable laser source (TLS) is tuned in time, the laser splits into two arms after 14 passing through the OC0. Then the frequency tuning laser launched into the main interferometer splits 15 into two portions at OC1: one (light a) enters the CIR, the PC1 and the FUT, the reference light (light 16 b) enters the PC2. The spontaneous Rayleigh backscattering light in the FUT interferes with light b at 17 the OC2. The interference light transmits to the PBS and then splits into two orthogonal polarization 18 components (P and S). P and S are detected by the photo detectors (PDs) and stored by the data 19 acquisition card (DAQ). In this way, polarization has no effect on the value of Rayleigh backscattering 20 amplitude. Meanwhile, the laser launched into the auxiliary interferometer splits into light c and light 21 d, an interference between them occurs at the OC4, the interference light is employed as a trigger signal 22 to mitigate the tuning nonlinearity.



TLS: Tunable Laser SourceOC: Optical CouplerCIR: Optical CirculatorPC: Polarization ControllerFUT: Fibre Under TestPBS: Polarization Beam SplitterPD: Photo DetectorDAQ: Data Acquisition Card



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Figure 1: (a) Configuration of a Coherent Optical Frequency Domain Reflectometry (C OFDR) system. (b) Interference between Rayleigh backscattering light and reference light.

By analysing the interference signal of the main interferometer, it can be found that a specific frequency f_B , named as the beat frequency, directly corresponds to a specific position (*z*) along the FUT, as indicated in Figure 1(b). The spatial resolution Δz can be given as

(1)

7
$$\Delta z = \frac{c}{2n_a \Delta F}$$

8 where c is the speed of light in space, n_a is the group refractive index, ΔF is the frequency tuning range 9 of the TLS. In known conditions, a stable and unique fingerprint Rayleigh backscattered spectrum 10 (RBS) is acquired. The local RBS shifts in frequency when a variation in the environmental conditions 11 occurs. The cross-correlation of the measurement RBS and the fingerprint RBS determines the local 12 spectral shift Δv , which can reveal local information (e.g. temperature and strain) along the whole fibre 13 by using proper calibration constants. A change in temperature or strain from the baseline condition 14 results in a frequency shift in the spectrum of light scattered in the fibre. Changes in the local period of 15 Rayleigh scattering cause temporal and spectral shifts in the locally reflected spectrum, which can be 16 scaled to form a distributed sensor. The strain response arises due to both the physical elongation and

compression of the sensor, and the change in fibre index due to photoelastic effects [26]. The thermal
 response arises due to the inherent thermal expansion of the fibre material and the temperature
 dependence of the refractive index [26].

The physical length and refraction index of the fibre are intrinsically sensitive to the measurement of both temperature and strain; therefore, the bare fibre sensor is used as the ε -DOFS. The ε -DOFS measures both strain and temperature simultaneously. The resultant measurement is solely a function of variations in temperature. As a result, this fibre is used as *T*-DOFS. The *T*-DOFS measures temperature only. Therefore, in order to capture distributed strain and temperature information respectively, a pair of ε -DOFS and *T*-DOFS were placed parallel and side by side to the subject under test, followed by temperature compensation of strain calibration.

11 The shift in the spectrum of light scattered in the ε -DOFS in response to strain and temperature is 12 analogous to a shift in the spectral shift, $\Delta v_{\varepsilon-DFOS}$:

13
$$-\frac{\Delta v_{\varepsilon-DFOS}}{v} = K_T \Delta T_{measured} + K_{\varepsilon} \varepsilon_{measured}$$
(2)

14 Where ν is the mean optical frequency, K_T and K_{ε} the temperature and strain calibration constants, 15 $\Delta T_{measured}$ and $\varepsilon_{measured}$ are the measured temperature change and strain.

16 This thermal response arises due to the inherent thermal expansion of the fibre material and the 17 temperature dependence of the refractive index. In the absence of any mechanical strain, the frequency 18 shift, Δv_{T-DFOS} , due to temperature change ΔT is defined as:

19
$$-\frac{\Delta v_{T-DFOS}}{v} = K_T \Delta T_{measured}$$
(3)

The default values for these constants are set at common values for germanosilicate core fibres: $K_T = 6.45 \times 10^{-6} \, ^{\circ}\text{C}^{-1}$ and $K_{\varepsilon} = 0.780 \, [27,28]$. The values for K_T and K_{ε} are dependent on the dopant species and concentration in the core of the fibre, but also to a lesser extent on the composition of the cladding and coating. Therefore, in the absence of strain, the temperature change can be defined as:

24
$$\Delta T_{measured} = -\frac{1}{K_T \nu} \Delta \nu_{T-DFOS} = C_T \Delta \nu_{T-DFOS}$$
(4)

25 Accordingly, the strain measurement can be compensated and calculated as follows:

26
$$\varepsilon_{measured} = -\frac{\kappa_{\varepsilon}}{\nu} (\Delta \nu_{\varepsilon - DFOS} - \Delta \nu_{T - DFOS}) = C_{\varepsilon} (\Delta \nu_{\varepsilon - DFOS} - \Delta \nu_{T - DFOS})$$
(5)

Assuming a scan centre wavelength of 1550 nm, the constants can be substituted in to yield the following temperature and strain conversion factors, respectively:

 $C_T = -0.801^{\circ}C/GHz$

$$C_{\varepsilon} = -6.67 \varepsilon \mu / GHz$$

1 **3 Methodology**

2 **3.1 Manufacturing of smart thermoplastic composite beam**

- 3 The process used to produce the thermoplastic composite beams with integrated DOFS involved three
- 4 steps, as shown in Figure 2.



<u>Preform preparation</u> <u>Embedding optical fibre</u>



Moulding in RVM press



5

6 Figure 2: Key steps in manufacturing thermoplastic braided beams. Left: Braiding onto 7 mandrel. Centre: Preparation for RVM and embedment of optical fibre. Right: Preform inside 8 the press. The red line highlighted the insertion of ε -DOFS.

9 The process steps were described in detail below:

Braiding and preform preparation: A commingled material system procured from Coats plc
 composed of E-glass fibres and polyamide 6 (PA6) filaments was used to braid the preforms.

12 The specification of the commingled material is listed in Table 2.

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Table 2 Properties of commingled E-glass/PA6 material.

Property	Value
Linear weight	1650 g/km
Fibre weight fraction	72%
Theoretical fibre volume fraction	54%
E-glass density	2.54 g/ml
PA6 density	1.15 g/ml
PA6 melting point	221 °C
PA6 glass transition temperature	55 °C

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The braiding of the circular hollow preforms was achieved at Composite Braiding Ltd using a 64-carier braiding machine. During the braiding process, the machine was operated at halfcapacity, i.e. 32 carriers were employed to result in a 1×1 or diamond braid architecture. Three layers were overbraided on an acetal mandrel with an outer diameter of 25 mm. The machine parameters were set to achieve a target braid angle of 45° in all the braided layers. Following the braiding process, the overbraided mandrels were transported to WMG for the moulding
 process.

- 3 ii. Integration of optical fibre: Preforms of approximately 480 mm long were extracted from the 4 received mandrels. The preform length was back calculated based on the final beam length as 5 well as the expected contraction during the moulding process. In order to secure the fibre 6 architecture during handling as well as moulding, both ends of the preform were taped using a 7 high temperature masking tape. The optical fibre employed in this study is a polyimide coated, 8 low bend loss, single mode fibre, which was manually integrated between the outermost and 9 second layer of the braided preform. This was performed by carefully removing the outer layer 10 and laying the optical fibre upon the second layer. In order to protect the optical fibre, a 0.6 mm 11 diameter Poly-Tetra-Fluoro-Ethylene (PTFE) coating was used in the ingress and egress 12 portions. The coated portions were also secured to the ends of the second layer of the braid 13 using masking tape to maintain orientation of the optical fibre. Finally, the outermost layer was 14 carefully reinstated, and a silicone rubber bladder was put through the preform. Thus, there 15 were no additional measures used to bond the optical fibre within the materials. Bespoke 16 designed metallic end fittings were attached at both ends of the bladder that would enable 17 internal pressurisation during the moulding process.
- *Moulding of braided preform*: The preforms were moulded using a novel bladder moulding
 technique called rapid variothermal moulding (RVM) developed previously by the authors [29].
 The process used a hydraulic press integrated with a modern smart tooling concept developed
 by Surface Generation [30]. The tooling setup enabled rapid heating/cooling of the mould faces
 while maintaining thermal uniformity across the mould via localised thermal control. The
 moulding process involved the following steps:
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a. Preheating to 150 °C followed by preform insertion and mould closure.

- b. Application of 4 bar bladder pressure via an external hose connected to the end fittings.
- c. Temperature increase to a consolidation temperature of 240 °C (20 °C above the melting point of PA6).

Rapid but uniform cool down to below 185 °C, i.e. recrystallisation peak of PA6.

- 29 d. A pressure increase to 20 bar as the consolidation temperature was reached.
- 30 e. Consolidation at 240 °C and 20 bar pressure for 15 minutes.
- 31

f.

32 g. Pressure release and demoulding of the consolidated preform at approximately 90 °C.

33 The produced tubular braided beam had an outer diameter of 35 mm. During the manufacturing process,

- 34 the Optical Frequency Domain Reflectometry (OFDR) system from the LUNA Inc. (ODiSI-B model)
- 35 was connected to the optical fibre and used to monitor the strain evolution in the composites. The total

1 strain from the interrogator was recorded and recalculated to account for the temperature compensation

2 of fibre thermal output.

3 3.2 Interrupted mechanical tests

4 A static three-point flexural test were conducted using an Instron 3367 test machine with a 30 kN load 5 cell in compression mode. The test setup was shown in Figure 3(a). The support span was 350 mm, 6 which was consistent with those tested in previous studies by the authors [31,32]. The tests were 7 performed in a loading-unloading interrupted scheme with a crosshead speed of 0.5 mm/min under 8 displacement control. The purpose behind the interrupted test was to assess the ability of the DOFS for 9 detecting damage in the tested composite beams with repetitive loads. In order to validate the 10 measurement from DOFS, the acquired strain data was compared with the surface strain distribution of 11 the braided beams captured using a three-dimensional digital image correlation (GOM 12 3D-DIC) 12 system [31,32] in previous studies. The test specimen used in this work was manufactured using the 13 same manufacturing parameters as the beams tested in the abovementioned previous studies. Therefore, 14 it was deemed reasonable to assume identical mechanical behaviour for the specimen manufactured and 15 tested in this work.

- During the interrupted tests, a travelling microscope equipped with a digital camera was used to monitor the damage near the central point of load application on the beam, as shown in Figure 3(b). The test scheme included six loading-unloading cycles with progressively increasing maximum displacement, as indicated in Table 3. The scheme was based on observations from previously conducted continuous tests on identical beams [31,32]. The tests were finally stopped at a crosshead displacement of 13 mm as significant lateral movement was observed at the beam-support roller interfaces upon deflections
- beyond this.



23 24



Loading cycle	Maximum displacement (mm)	Maximum load (N)
1	3.0	272.96
2	4.5	360.90
3	6.0	427.46
4	7.5	465.77
5	9.0	471.38
6	10.5	447.94

Table 3 Loading-unloading cycles of interrupted three-point bending test.

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The DOFS was setup to acquire data at a rate of 1 Hz with a gauge length of 1.25 mm. Before starting any measurement, a reference OF tare was set up to erase any residual strains from the manufacturing process and measure solely the strain induced due to the application of mechanical load

6 3.3 Micro-CT

7 An X-Ray micro Computed Tomography (micro-CT) scan was performed at the Centre for Imaging, 8 Metrology and Additive Technologies (CiMAT) at WMG, Warwick University to extract detailed 9 information regarding the orientation and positioning of the optical fibre within the three-dimensional 10 volume of the specimen, more generally, the quality of the embedment of the optical fibre sensor in the 11 material. In this work, a TESCAN UniTOM XL system was used, while the post analysis was performed 12 using commercially available software VGStudio Max 2.2.5. The obtained resolution of the X-Ray 13 micro-CT scan was 35 µm and has been chosen as a compromise between a good image quality and 14 being able to see as long as the optical fibre presents as the optical fibre might not be in the same 15 visualised plane. The whole sample was scanned by merging 8 scans of region of interest. The other 16 parameters used in the micro-CT scanning are listed in Table 4. The images obtained from the micro-17 CT scanning will be discussed in Section 4.2.

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Table 4 Parameters used in the micro-CT scanning.

Exposure Voltage	80 kV
Exposure Power	35W
Exposure Time	75 msec
Voxel size	35 µm
No of Projections	1979
Frame per projections	3

19 4 Results and discussion

20 4.1 Strain monitoring during consolidation and cooling

21 Figure 4(a) displayed the temporal characteristics of the total strain shift along the embedded optical

22 fibre length (measured from ε -DOFS) and the mould temperature (obtained from the Surface Generation

user interface). Figure 4(b) showed the strain mapping along the embedded optical fibre length during

the consolidation and cooling process, with a clear distinction in the strain values across the phase transition (from liquid phase before recrystallisation to solid phase after recrystallisation). During the manufacturing process, the presence of rigid metallic end fittings at either end of the beam resulted in a physical constraint upon the optical fibre. This also resulted in the breakage of the optical fibre in close vicinity to the end fittings in several manufacturing trials. Therefore, in order to eliminate any artifacts introduced due to the end fittings, the strain data within approximately 150 mm from both the beam ends was omitted.

9 The temporal strain distribution was divided into four distinct stages (numbered I to IV) corresponding 10 to the moulding process as described in Section 3.1. During the process timeline shown in Figure 4(a), 11 the preform was always under a bladder pressure of at least 4 bar. Therefore, the strains represented and 12 discussed here are a combination of thermal and mechanical strains. Moreover, for every stage, lengthwise strain data was also analysed. The standard deviation in the measured strain corresponding 13 14 to a selection of temperature from different stages were listed in Table 5. Moreover, several snapshots 15 depicting these variations were also shown with the temporal strain and temperature distribution in 16 Figure 4(c).



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Figure 4: (a) The evolution of strain range along the embedded optical fibre length and the temperature measurement, (b) snapshots showing strain profiles along the length of the beam during different stages of the manufacturing process and (c) strain mapping along the embedded optical fibre length during the consolidation and cooling process.



Table 5 Standard deviation in the measured strain profiles during different stages of the manufacturing process.

S.No.	Temperature (°C)	Stage	Standard deviation (με)
1	170.98	Ι	191.380
2	223.49	II-before pressure application	129.279
3	240.08	II-after pressure application	121.232
4	240.53	III- during consolidation	126.744
5	201.73	IV- before recrystallisation	198.593
6	161.88	IV-post recrystallisation	264.951

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12 The four stages of strain evolution are discussed below: I. The mould ramped up to the consolidation temperature (240 °C) during the first stage. The temperature rise was primarily responsible for the observed positive strain. As shown in Table , the standard deviation during the initial phase of stage I were comparatively higher than the subsequent stages when the matrix was in a molten state. This is because before the PA6 melted, the material was in the form of bulkier commingled glass and PA6 tows. Therefore, the optical fibre underwent relatively severe undulations along its length compared to later stages after the melting of the PA6. This was schematically represented in Figure 5.

8 II. In stage II, the PA6 was already above its melt temperature of 221 °C. Upon reaching the 9 consolidation temperature of 240 °C, the pressure was increased from 4 to 20 bar. This increase 10 in pressure was accompanied with a sharp drop in the strain values. This sharp drop was a result 11 of the higher internal pressure upon the braided tows, which resulted in outwards squeeze of the 12 molten PA6 from the tows, thus compressing the tows in the thickness direction. This 13 phenomenon was also depicted in Figure 5. As shown in the figure, the decrease in the tow 14 thickness reduced the undulations in the optical fibre, thus resulting in a decrease in tension along 15 their length.



Figure 5: Schematic representation of decrease in optical fibre undulation upon reduction inbraided tow thickness.

- III. During stage III, a temperature of 240 °C was maintained for 15 minutes, The strain values
 showed a stable trend during this regime with a constant peak to peak strain value as listed in
 Table .
- IV. The cooling began in this stage and the strains started to decrease with the temperature. It could
 be observed from Table that the deviation in the strain values increased with decreasing
 temperature. The rise was particularly higher beyond the recrystallisation temperature of 185 °C,
 i.e., the point at which the matrix solidified. This effect was further elaborated in Figure 6, which
 shows the lengthwise strain evolution during the cooling phase of the process.
- 27







3 This rise could be attributed to two competing mechanisms. Firstly, as reported for the bladder moulding 4 of the considered material system previously [32], the glass fibres on the outer surface showed waviness. 5 This distortion was speculated to be caused by the compressive residual strains induced during the 6 cooling phase of the process. Similar results were also reported in [33]. The optical fibre would also 7 experience comparative residual strains in along its axis and hence undergo waviness. A Micro-CT 8 scanning image of the consolidated beam, as shown in Figure 7, confirmed the presence of waviness in 9 the optical fibre. Therefore, the resulting variability in the orientation of the optical fibre would cause 10 the measured strain to deviate from the braided composite beam's axis and therefore induce a variation 11 in the results.

Secondly, during the post-consolidation, the braided composite had a periodic inhomogeneous material system, which is inherent to all textile composites. In the context of thermal strains, this inhomogeneity manifested in the form of variable thermal expansion coefficient, i.e., the optical fibre was subjected to periodically repetitive variations in thermal expansion contraction as per the local thermal properties.

Eventually, a mean compressive residual strain of 479 με was recorded before demoulding the beam at
approximately 90 °C.

18 **4.2 Profile of DOFS in braided composites**

As the DOFS was embedded in braided composites, the differences in compaction of the yarns/tows
 during resin consolidation may lead to different sensor profiles. Therefore, micro-CT was used as a tool

- 21 to help interpret the position check of the sensor profile. **Error! Reference source not found.** showed
- 22 the non-planar view of two sections extracted from the 3D rendering. Three images were presented: the

1 first two images (a, b) showed the embedded DOFS between the layers of braided composites, (c) is an 2 enlarged image of part at the termination of the composite beam where the fibre was encapsulated within 3 a protective PTFE coating and (d) is the side view of the enlarged embedded optical fibre within 4 composites. All the images indicated negligible control over the orientation of the optical fibre within 5 the braided composite. As discussed in Section 4.1, there were multiple phenomenons that could result 6 in an undulating/wavy profile of the optical fibre. It was also observed in [24] where the OF was 7 embedded between woven-fabric (90,0) layers and a shift of OF profile between the adjacent woven 8 plies was observed as well as some differences in the spectrum of the OF reflected signal (e.g., peak 9 splitting and shouldering). In reference [10], micro-braiding of the DOFS with a glass tow was 10 attempted for improved handling and mechanical properties of the OFs, however, the micro-braiding 11 did not affect the sensitivity. Nevertheless, it was found in this study that the strain measurements were comparable and reasonable for the investigation, however, it would also be interesting to introduce the 12 13 micro-CT images to identify the OF profile to understand the standing switch during the manufacturing. 14 As the signals did not show spikes despite waviness, it was believed that the orientation was not crucial 15 for the purpose of SHM.



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Figure 7: Non-planar view of two sections (*a*, *b*) taken along the beam embedded with DOFS.
The highlighted red line indicated the locations of the optical fibre; (c) is the enlarged image
taken from image *b* to show the termination layout of the bare DFOS and bare DFOS with
PTFE coating; (d) is the side view of the enlarged embedded optical fibre within composites

21 **4.3** Structural health monitoring during three-point flexure test

22 The load-midspan displacement curve obtained from interrupted tests was presented in Figure 8. Every

23 loading cycle (LC) was composed of two phases: a rising load phase (RL) and a decreasing load phase

- 24 (DL). In the initial load cycle, i.e. LC1, the load-displacement data showed a linear trend during the RL
- 25 phase. However, during the DL phase, an offset relative to the RL phase was observed. During the RL

1 phase of LC2, the load replicated the linear curve obtained from RL during LC1. This overlap in the 2 load-displacement curve indicated that the material retained its stiffness and therefore did not undergo 3 any damage during LC1. The observed offset after load removal in DL phase (of LC1) could have 4 occurred due to a rate-dependent relaxation phenomenon, however, further works needed to confirm 5 this speculation. After moving beyond the load threshold of LC1, there was observable non-linearity in 6 the LC2 curve. In addition, upon the end of the DL phase in LC2, the positional offset from the initial 7 point increased further. In the next loading cycle LC3, the RL phase showed a decrease in stiffness 8 compared with LC1 and LC2. These observations indicated that the non-linearity observed in LC3 could 9 be a sign of microcracks damage initiation ($< 50 \mu m$). As expected, the stiffness degradation and 10 increase in zero-load offset were observed to rise in the further load cycles (LC3 to LC4). The failure 11 load was kept increasing with the gradual reduction on the material stiffness, which might be related to 12 the generation of fictitious cracks (> 50 μ m). Finally, when the beam resistance was maximum with the

- 13 force of 471.3 N (LC5), gradual degradation with a rise in deflection continued as the localised fictitious
- 14 cracks propagated and the fibre was yielding, with a comparatively reduced peak load in LC6.



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Figure 8: Load-displacement curve of the braided composite beam during interrupted quasistatic test. The load cycles (LCs) in interrupted tests are marked. The left corner image indicates the virtual measurement line of DIC.

Figure 9 (left) showed the horizontal strain distribution measured using the DIC along the central portion (depth-wise). Despite the minute undulations, the CT images from Figure 7 indicated that the fibre was relatively straight in the central portion of the beam, where DIC strain measurements were performed. Hence, this comparison is suitable. For DIC, the virtual measurement lines (indicated in Figure 9, positioned at 10.9 mm above the centre line), which was at the same height as the location of DOFS were prepared. The comparisons between DIC data and DOFS measurements at the

1 displacements corresponding to maximum displacements in LC1 to LC3 were used for validation. For 2 LC1 to LC3, a reasonable correlation was observed between the DIC and DOFS data. This is because, 3 despite the interrupted nature of the performed tests in this work, the beam did not undergo any damage 4 in LC1 and hence the LC2 deformation was identical to that of a beam tested continuously. Despite the 5 stiffness drop observed in LC3, the strain data from DIC and DOFS showed similar trends, with an 6 observable positive strain developing in the region corresponding to the load application point. As noted 7 in references [31,32], this was likely due to the localised loading underneath the roller that resulted in 8 a deviation from the strain profile observed for ideal three-point flexure.



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Figure 9: Strains measured along the embedded optical fibre length under cycle 1 to cycle 3
(left) compared with DIC data (right) for validation only, the black and red lines are for DIC
and DFDS, respectively; the colour scale for DIC is presented.

13 Figure 10(a) showed the strain profiles taken at the maximum displacement points of each load cycle 14 along the embedded optical fibre length. The increase in the strain peak from LC1 to LC3 was constant 15 until a significant strain change occurred at LC4. At LC4, localised peaks around the middle position 16 were visible from DOFS: this abrupt change could be an indication of significant local damage within 17 fibres. At LC5, some dropout points started to appear at around crack locations, which was caused by 18 the spectral shift calculation algorithm due to the low correlation with the reference spectrum [17]. This 19 phenomenon of miscalculated spike point was increasing due to the rapid variation in the strain gradient 20 in composites for a 1.25 mm gage length in the ODiSI-B system, which was also observed in [10,17]. 21 During the loading and unloading associated with LC1 to LC5, the strain data recorded by DOFS was 22 able to return to the initial starting point during the unloading process. However, for LC6, the strain 23 configuration measurement indicated the breakage of the optical fibre so the signal could not return. As

1 the strain profile showed an undulating profile, a Fast Fourier Transform (FFT) study was conducted to 2 convert the original signal to a representation in a hypothetical frequency domain in order to further 3 differentiate between the strain profiles pertaining to different cycles. The normalised FFT data, shown 4 in Figure 10 (b) and Figure 10(c), indicated no pattern shift in the first three LCs. However, a significant 5 variation was observed for LC4, which could be associated with a significant damage initiation. In order 6 to validate this, surface pictures of the loaded region of the beam captured using the travelling 7 microscope at 40X magnification were observed. These pictures were shown in Figure 11. The pictures 8 were taken after the unloading process of each cycle. While there was no visible sign of surface cracks 9 till LC3, a dominant crack and a side crack were suddenly observed since LC4. These cracks were 10 further observed to propagate during LC5. Based on the position with respect to the embedded length 11 of the DOFS, it was speculated that the dominant crack propagated to the DOFS location during LC6, 12 which resulted in the breakage of DOFS. Therefore, the post observation results and the correlations 13 between visible damage and a clear change in the strain signal after FFT processing have proved that 14 the embedded DOFS was sensitive to the growth of damage to a certain length in the composites and 15 could be implemented for structural health monitoring. Although it is difficult to locate the damage 16 initiation second due to the lack of efficient damage observation methodology and a schematic un-17 supervised data analysis of the strain data, the results were still significant as it could capture the instant 18 when considerable damage exist (i.e., approximately 5 mm) for inspection check upon damage tolerance 19 strategy.





Figure 10: (a) Strain profiles along the embedded optical fibre length at the maximum
displacement points of different load cycles, (b) The normalised FFT study of LC1 to LC3 and
(c) LC1 to LC4.



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Figure 11: Damage observation after the unloading process of different cycles (a) cycle 3 (6
mm), (b) cycle 4 (7.5 mm), (c) cycle 5 (9 mm), (d) cycle 6 (10.5 mm) taken by 40× travelling
microscope. The crack was marked by the red line.

9 The above results indicated that despite the lack of control over the optical fibre's orientation, by 10 processing and comparing the acquired strain values over time allowed for identification of damage. 11 This showed that the crude nature of the method adopted to embedded optical fibre within the 12 composite did not affect its ability to monitor damage and successfully performed the expected role 13 on structural health monitoring. Thus, without adding significant cost or time to the composite 14 manufacturing process, the results demonstrated a simple yet effective route towards SHM of braided 15 composite structures.

16 **5 Conclusions**

17 This paper demonstrated the integration of DOFS into a braided thermoplastic composite component. 18 The optical fibre was manually embedded between the two outermost layers of a braided composite 19 beam such that the manufacturing process was barely disrupted. The ODiSI-B system coupled with the 20 embedded DOFS was shown to successfully measure strains during the consolidation process as well 1 during mechanical testing in real-time. The position of OF profile was further interpreted by a micro-

2 CT scanning to define a reliable embedding.

3 The strain measured during the moulding process was correlated with the thermal and mechanical 4 changed that occurred in the material. The measured strain of the braided material was directly 5 correlated with the temperature trend at every stage of the consolidation and cooling process, shifting 6 from tensile strain to compressive strain during the cooling due to the stress relief of the resin. The 7 phase change from a molten state to semi-crystalline could be observed from strain measurement. 8 Following the demoulding at 90 °C, the residual strain was measured, which was difficult and expensive 9 to evaluate from the traditional approaches. The observed patterns in the strain measurements along the 10 embedded optical fibre length during the cooling stage showed an undulating trend, which increased 11 with reducing temperature. This was attributed to two phenomenons: (i) waviness caused in the optical fibre as a consequence of thermal contraction or lack of orientation control during the embedding 12 13 process or (ii) the periodic braid architecture of the fully consolidated part with a varying thermal 14 property along the length of the beam.

The micro-CT scans of the consolidated braided beam revealed a lack of consistency in the optical fibre's orientation. Moreover, they also revealed the presence of waviness in the embedded optical fibre, thus validating its effect on the observed increase in the undulating strain measurements. Further research on integrating the DOFS during braiding as tri-axial preforms for strain measurement would be significant.

20 The SHM capability of the DOFS was assessed by performing three-point flexure tests on the 21 manufactured beam. The validity of the DOFS's strain measurement was confirmed through a 22 reasonable correlation between the obtained strain values and surface strain data from DIC 23 measurement. In order to track the damage of the beam, a travelling microscope was used to record 24 pictures of the loaded region after each unloading stage. The obtained strain data was further processed 25 using a Fast Fourier Transform. It was found that an abrupt shift in the processed data had a direct match 26 with a significant damage observed via the captured surface pictures, thus proving the effectiveness of 27 the DOFS in indicating damage for inspection check.

The results proved that the approach used to integrate the optical fibre without any specific mechanism to ensure its orientation (hence cause minimum disruption to the manufacturing process) was a valid method for performing structural health monitoring. Despite the lack of orientation control and induced imperfections in the optical fibre during the moulding process, the comparative analysis of the strain signal after being processed via FFT showed a significant shift corresponding to the load cycles in which visible damage was observed.

Further work aimed at repeating the experiments with reduced gap between consecutive load cycles
would establish the absolute limit of the optical fibre's damage detection ability. Moreover, in-situ NDT

1 (i.e., Computed Tomography during manufacturing as well as mechanical testing) could provide further

2 insights into the changes occurring in the composite and correlate them with strain measurements for

- 3 improving the capability of optical fibre to assess manufacturing defects as well as damage initiation in
- 4 real-life structures. Moreover, further research based on data-driven models is also recommended as a
- 5 means of creating more efficient methodologies to produce reliable un-supervised damage estimation.

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12 7 Data availability statement

13 The datasets generated during and/or analysed during the current study are available from the 14 corresponding author on reasonable request.

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